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# The Crystal Structure of $\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$, a Fluorite-Related Structure Containing Rhenium Doublets 

By Kjell Waltersson<br>Department of Inorganic Chemistry, Arrhenius Laboratory, University of Stockholm, Fack, S-104 05 Stockholm, Sweden

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#### Abstract

The crystal structure of $\mathrm{La}_{2} \mathrm{ReO}_{5}$ has been determined from three-dimensional X-ray diffractometer data (Mo $K \alpha$ radiation); the constitutional formula was found to be $\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$. The tetragonal unit cell, space group $I 4 / m$ (No. 87), has the dimensions $a=8.935, c=6.011 \AA, V=479.9 \AA^{3}$. The unit cell contains two formula units of $\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$. Least-squares refinement reached $R_{w}=0.0438$ for 2223 measured independent reflexions for which $\sigma\left(F_{o}\right) / F_{o} \leq 0 \cdot 20$. The structure is closely related to the fluorite type. The O atoms are in the F positions while four of five Ca sites are occupied by La atoms and the fifth accommodates a pair of Re atoms with its axis parallel to the $c$ axis. The $\mathrm{Re}-\mathrm{Re}$ distance is only 2.259 (1) $\AA$, close to the distance reported for $\mathrm{K}_{2} \mathrm{Re}_{2} \mathrm{Cl}_{8} .2 \mathrm{H}_{2} \mathrm{O}$. The structural relationships between $\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$ and $\mathrm{Nd}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{11}$ are discussed.


## Introduction

Although the preparation of ternary oxides containing rare-earth elements including Y and $\mathrm{La}(=\mathrm{Ln})$ and Re in different oxidation states has been reported by several authors (see below) the structures of only a few of these compounds have been reported, viz. $\mathrm{La}_{4} \mathrm{Re}_{6} \mathrm{O}_{19}$ (Morrow \& Katz, 1968; Longo \& Sleight, 1968) and $\mathrm{Nd}_{4} \mathrm{Re}_{2} \mathrm{O}_{11}$ (Wilhelmi, Lagervall \& Muller, 1970).

Extensive preparative studies of the Ln metal-rich ( $\mathrm{Ln} / \mathrm{Re} \sim 2$ ) part of several lanthanoid rhenium-oxide systems were reported by Muller (1968), Muller \& Roy (1969), and Baud \& Capestan (1968). Five structure types with nearly the same stoichiometry, $\mathrm{Ln}_{2} \mathrm{ReO}_{5}$, were reported by Muller \& Roy (1969), viz.
$\beta-\mathrm{Nd}_{2} \mathrm{ReO}_{5.25}$ type ( $\mathrm{Ln}=\mathrm{Nd}, \mathrm{Sm}$ and Gd ), $\alpha-\mathrm{Nd}_{2} \mathrm{ReO}_{5}$ type ( $\mathrm{Ln}=\mathrm{La}$ and Nd ), $\alpha-\mathrm{Sm}_{2} \mathrm{ReO}_{5}$ type ( $\mathrm{Ln}=\mathrm{Sm}$ and $\mathrm{Gd}), \mathrm{Yb}_{2} \mathrm{ReO}_{5}$ type ( $\mathrm{Ln}=\mathrm{Gd}, \mathrm{Dy}, \mathrm{Y}$, Er and Yb ) and the $\mathrm{Y}_{2+x} \mathrm{ReO}_{5+y}$ type ( $\mathrm{Ln}=\mathrm{Y}$ ).

The Nd-compound, belonging to the first structure type, was investigated some years ago by Wilhelmi et al. (1970). In view of the structure determination, the compound was reformulated as $\mathrm{Nd}_{4} \mathrm{Re}_{2} \mathrm{O}_{11}$ (or $\left.\mathrm{Nd}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{11}\right)$. This paper deals with the structure determination of $\mathrm{La}_{2} \mathrm{ReO}_{5}\left(\alpha-\mathrm{Nd}_{2} \mathrm{ReO}_{5}\right.$ type).

## Experimental

A black crystalline sample of the compound was prepared by Dr A. W. Sleight (Central Research Department, E. I. du Pont de Nemours and Co., Wilmington, Delaware 19898, U.S.A.). The selection of a suitable crystal and the determination of possible space groups were performed by X-ray photographic methods (oscillation and Weissenberg techniques). The crystal, $0.050 \times 0.050 \times 0.114 \mathrm{~mm}$, was mounted with the largest dimension parallel to the crystallographic $c$ axis. Cell dimensions were calculated from diffractometer measurements (Table $1 a$ ) and were the same as those reported by Muller \& Roy (1969), within the limits of error.

The X-ray intensity data were collected by a Siemens AED diffractometer with graphite-monochromatized Mo $K \alpha$ radiation and a scintillation counter with pulseheight discrimination. The $\theta-2 \theta$ scan technique was used. Each reflexion was measured according to the five-values technique. The scan range for the individual reflexions was given by the expressions $+\Delta \theta=0.78+$ $0.12^{\circ} \tan \theta$, and $-\Delta \theta=0 \cdot 70+0 \cdot 17^{\circ} \tan \theta$. The constants were evaluated after scanning a few reflexions manually and recording the widths of the peaks. Each
intensity measurement was taken in steps of $0.01^{\circ}$ in $\theta$, with the maximum step time 0.60 s . For strong reflexions the step time was automatically changed in order to accelerate the data collection. In order to keep

Table $1(a)$. The crystal structure of $\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$
$\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$ : M.W. 1088-0. Space group: $14 / \mathrm{m}$ (No. 87). Cell dimensions according to Muller \& Roy (1969): $a=8.935$, $c=6.011 \AA$ and $V=479.9 \AA^{3} . D_{c}=7 \cdot 42, D_{o}=7.53 \mathrm{~g} \mathrm{~cm}^{-3}$ (Baud \& Capestan, 1968). Cell content: $2 \mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$.

|  | Point | set | Numberin | of atoms |
| :---: | :---: | :---: | :---: | :---: |
|  | 4(d) | $\frac{1}{2}, 0,4$ | ( $n$, |  |
|  |  | $0, \frac{1}{2}, \frac{1}{4}$ | $(n$, |  |
|  |  | $0, \frac{1}{2}$, $\frac{3}{4}$ | $(n$, |  |
|  |  | $\frac{1}{2}, 0, \frac{3}{4}$ | $(n$, |  |
|  | 4(e) | 0, $0, z$ | (1) |  |
|  |  | $0,0, \bar{z}$ | (2) |  |
|  |  | $\frac{1}{2}, \frac{1}{2}, \frac{1}{2}+z$ | (3) |  |
|  |  | $\frac{1}{2}, \frac{1}{2}, \frac{1}{2}-z$ | (4) |  |
|  | $8(h)$ | $x, y, 0$ | (1) |  |
|  |  | $\bar{x}, \bar{y}, 0$ | (2) |  |
|  |  | $\bar{y}, x, 0$ | (3) |  |
|  |  | $y, \bar{x}, 0$ | (4) |  |
|  |  | $\frac{1}{2}+x, \frac{1}{2}+y, \frac{1}{2}$ | (5) |  |
|  |  | $\frac{1}{2}-x, \frac{1}{2}-y, \frac{1}{2}$ | (6) |  |
|  |  | $\frac{1}{2}-y, \frac{1}{2}+x, \frac{1}{2}$ | (7) |  |
|  |  | $\frac{1}{2}+y, \frac{1}{2}-x, \frac{1}{2}$ | (8) |  |
|  | 16(i) | $x, y, z$ |  |  |
|  |  | $\bar{x}, \bar{y}, z$ |  |  |
|  |  | $x, y, z$ |  |  |
|  |  | $\bar{x}, \bar{y}, \bar{z}$ |  |  |
|  |  | $\bar{y}, x, z$ |  |  |
|  |  | $\bar{y}, \bar{x}, z$ |  |  |
|  |  | $\bar{y}, x, \bar{z}$ |  |  |
|  |  | $y, \bar{x}, \bar{z}$ |  |  |
|  |  |  |  |  |
|  |  | $\frac{1}{2}-x, \frac{1}{2}-y, \frac{1}{2}+z$ |  |  |
|  |  | $\frac{1}{2}+x, \frac{1}{2}+y, \frac{1}{2}-z$ |  |  |
|  |  | $\frac{1}{2}-x, \frac{1}{2}-y, \frac{1}{2}-z$ |  |  |
|  |  | $\frac{1}{2}-y, \frac{1}{2}+x, \frac{1}{2}+z$ |  |  |
|  |  | $\frac{1}{2}+y, \frac{1}{2}-x, \frac{1}{2}+z$ |  |  |
|  |  | $\frac{1}{2}-y, \frac{1}{2}+x, \frac{1}{2}-z$ |  |  |
|  |  | $\frac{1}{2}+y, \frac{1}{2}-x, \frac{1}{2}-z$ |  |  |
| Atom( $n$ ) | Point set | $x+\sigma(x)$ | $y+\sigma(y)$ | $z+\sigma(z)$ |
| Re | 4(e) | 0 0 | 0 | $0 \cdot 18788$ (4) |
| La | $8(h)$ | $0 \cdot 18154$ (3) 0 | $0 \cdot 39372$ (3) | 0 |
| O(1) | 4(d) | $\frac{1}{2} 0$ | 0 | $\frac{1}{4}$ |
| $\mathrm{O}(2)$ | 16(i) | 0.09517 (32) 0. | -18615 (30) | $0 \cdot 25775$ (54) |

counting losses low for strong reflexions, the automatic attenuator mechanism was set to add filters.

Two standard reflexions (004 and 060), automatically monitored throughout the data collection, indicated no deterioration of the crystal.
All the 2337 possible reflexions, systematic extinctions excluded, within one octant of the reciprocal lattice sphere to $\theta=65^{\circ}$, were measured. The 2223 reflexions with $\sigma\left(I_{n}\right) / I_{n} \leq 0.40$ were used in the subsequent calculations. The net intensity $I_{n}$, was calculated as $I_{t}-I_{b}$, and its estimated standard deviation was obtained from the expression $\sigma\left(I_{n}\right)=\left(I_{t}+I_{b}\right)^{1 / 2}$, where $I_{t}$ and $I_{b}$ are the number of counts for the total and background intensities respectively.

The net intensities were corrected for Lorentz, polarization and absorption effects. The linear absorption coefficient, $\mu(\mathrm{Mo} K \alpha)=441 \mathrm{~cm}^{-1}$, was used in calculating a transmission factor for each reflexion (International Tables for X-ray Crystallography, 1962). The absorption was estimated by the Gaussian integration method with $6 \times 6 \times 16$ grid-points. The factor $(A)$ in the formula $I=\operatorname{Lp} . I_{n} / A$ ranged from $0 \cdot 1475$ to $0 \cdot 2279$. Preparatory calculations for extinction correction were included.

The calculations were performed by IBM 1800 and IBM 360/75 computers.

## Structure determination and refinement

The crystal structure of $\mathrm{La}_{2} \mathrm{ReO}_{5}$ belongs to the tetragonal system with the reflexions $h+k+l=2 n+1$ systematically absent; the possible space groups were: $I 4 / m$ (No. 87), $I 4$ (No. 79 ) and $I \overline{4}$ (No. 82).

The structure determination was started with the assumption that the space group $I 4 / m$ (No. 87) was the correct one. From a three-dimensional Patterson synthesis tentative coordinates of one Re atom were derived in point set $4(e)$. The discrepancy index $R=\Sigma| | F_{o} \mid-$ $\left|F_{c}\right| / / \Sigma\left|F_{o}\right|$ was at this stage $0 \cdot 28$.

From a three-dimensional ( $F_{o}-F_{\mathrm{Re}}$ ) synthesis with approximate signs given by the refined Re position, one La atom was located in point set $8(h)$. These metal positions explained all the prominent vectors in the

Table 1(b). Final anisotropic temperature factors defined as $\exp \left[-\left(\beta_{11} h^{2}+\beta_{22} k^{2}+\beta_{33} l^{2}+2 \beta_{12} h k+2 \beta_{13} h l+2 \beta_{23} k l\right)\right.$ ( $\beta_{l j}$ values are given in $\AA^{2} \times 10^{5}$, estimated standard deviations in parentheses) with r.m.s. components ( $R_{i}$ ) of ther mal displacements along the ellipsoid axes and the r.m.s. radial $(R)$ thermal displacements

| Atom ( $n$ ) | $\beta_{11}$ | $\beta_{22}$ | $\beta_{33}$ | $\beta_{12}$ | $\beta_{13}$ | $\beta_{23}$ | $R_{1}$ | $R_{2}$ | $R_{3}$ | $R$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Re | 94 (2) | 94 (2) | 220 (4) | 0 | 0 | 0 | 0.0617 | 0.0617 | 0.0635 | 0.1079 |
| La | 107 (2) | 142 (2) | 338 (4) | 6 (1) | 0 | 0 | 0.0655 | 0.0759 | 0.0787 | $0 \cdot 1275$ |
| $\mathrm{O}(1)$ | 166 (39) | 166 (39) | 152 (61) | 0 | 0 | 0 | 0.0527 | 0.0819 | 0.0819 | $0 \cdot 1272$ |
| $\mathrm{O}(2)$ | 170 (17) | 133 (16) | 546 (46) | -18(13) | -18(24) | 0 (23) | $0 \cdot 0713$ | 0.0842 | $0 \cdot 1003$ | $0 \cdot 1491$ |
| Extinction parameters $\times 10^{4}$ |  |  |  |  |  |  |  |  |  |  |
|  | $\begin{array}{r} \varepsilon_{1} \\ 12.37 \end{array}$ |  | $\begin{gathered} \varepsilon_{22} \\ 2 \cdot 63(24) \end{gathered}$ | $\begin{gathered} \varepsilon_{33} \\ 3 \cdot 80(39) \end{gathered}$ | $-1.00$ |  | $\begin{gathered} \varepsilon_{13} \\ 3 \cdot 14(50) \end{gathered}$ | $\varepsilon_{23}$ -1.85 |  |  |

Principal semi-axis of the assumed ellipsoidal particle (microns)

$$
0.202(10) \quad 0.438(20) \quad 0.365(20)
$$

three-dimensional Patterson map. Refinement of these positions yielded a conventional $R$ value of $0 \cdot 10$. Two $O$ atoms were next located, one in point set $4(d)$ and one in 16(i), from a three-dimensional difference ( $F_{o}-$ $F_{\text {metal }}$ ) synthesis, calculated from the refined metal coordinates. With the O positions included, least-squares refinement reduced $R$ to 0.08 .
Refinement was continued with a full-matrix program including anisotropic temperature factors and anisotropic extinction corrections. The scattering factors were taken for neutral atoms from McMaster, Kerr del Grande \& Hubbell (1969), and the anomalous scattering values were taken from International Tables for X-ray Crystallography (1962).

The quantity minimized in the least-squares refinement was $\sum w\left(\left|F_{o}\right|-\left|F_{c}\right|\right)^{2}$. The conventional $R$ value and the weighted $R_{w}$ value, defined by $R_{w}=\left(\sum w \Delta^{2}\right)^{1 / 2} /$ $\left(\sum w F_{o}^{2}\right)^{1 / 2}$, were 0.0351 and 0.0438 respectively for the final cycle of refinement. The weights applied to the structure factors were calculated according to the formula:

$$
w=\left[\sigma^{2}\left(F_{o}\right)+10+F_{o} / 80+80 / F_{o}\right]^{-1} .
$$



Fig. 1. The two different fluorite subcells. The La atoms or the Re pairs are situated either at the corners or at the faces of the subcells.

There were 27 variables in final least-squares refinement: the six positional parameters, 14 anisotropic temperature-factor coefficients, one scale factor, and six anisotropic-extinction parameters.

Anisotropic-extinction corrections were performed, with extinction assumed to be dominated by particle size (type II) (Coppens \& Hamilton, 1970). The assumed ellipsoidal particle shape is described by the tensor $\varepsilon_{i j}$.

The principal semi-axis $j$ of the ellipsoid is given by the expression $\lambda / / \varepsilon_{i j}$ (in microns) (see Table $1 b$ ). Corrections with extinction assumed to be dominated by mosaic spread (type I) (Coppens \& Hamilton, 1970) were also tested, but both the $R$ and $R_{w}$ values were significantly higher than for type II, which therefore was considered the more realistic model.

Attempts were also made to refine the structure in the space groups $I 4$ (No. 79) and $I \overline{4}$ (No. 82). The coordinates thus obtained did not differ significantly from those found for space group $I \overline{4} / m$ (No. 87), which was therefore considered as the actual symmetry of the compound.

The final positional, thermal, and extinction parameters together with their estimated standard deviations as well as the numerical values of the root mean square thermal displacements are given in Table 1(b). The weight

Table 2. Weight analysis obtained after the final cycle of the least-squares refinement of $\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$
$\Delta=\left|\left|F_{o}\right|-\left|F_{c}\right|\right|, w=$ weighting factor. The $\overline{w \Delta^{2}}$ values have been normalized.

| Interval <br> in $F_{o}$ | $w \Delta^{2}$ | Number of <br> reflexions | Interval <br> in $\sin \theta$ | $\overline{w \Delta^{2}}$Number of <br> reflexions |  |
| :---: | :---: | :---: | :---: | :---: | :---: |
| $0-20$ | 0.76 | 240 | $0.0-0.422$ | 2.77 | 264 |
| -30 | 1.22 | 253 | -0.532 | 0.99 | 246 |
| -40 | 0.66 | 220 | -0.609 | 2.74 | 233 |
| -50 | 0.96 | 217 | -0.670 | 0.54 | 226 |
| -60 | 0.64 | 199 | -0.722 | 0.52 | 225 |
| -75 | 0.67 | 251 | -0.768 | 0.35 | 220 |
| -90 | 0.89 | 178 | -0.808 | 0.49 | 201 |
| -115 | 0.45 | 240 | -0.845 | 0.35 | 225 |
| -150 | 2.62 | 199 | -0.879 | 0.32 | 208 |
| $>150$ | 1.30 | 225 | -0.906 | 0.30 | 174 |



Fig. 2. The two metal coordination polyhedra in the $\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$ structure. The atomic distances are given in $\AA$. The thermal ellipsoids are drawn at $90 \%$ probability. (a) The $\mathrm{Re}_{2} \mathrm{O}_{8}$ unit. (b) The $\mathrm{LaO}_{8}$ polyhedron.
analysis after the last cycle of least-squares refinement is shown in Table 2.* Figs. 1 and 2 are pictorial repre-

* A list of structure factors has been deposited with the British Library Lending Division as Supplementary Publication No. SUP 31377 ( 11 pp., 1 microfiche). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 13 White Friars, Chester CH1 1NZ, England.

Table 3. Interatomic distances $(\AA)$ and bond angles $\left({ }^{\circ}\right)$ in $\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$
Estimated standard deviations in the last decimal place are given within parentheses. The atoms are numbered according to Table $1(a)$.
(a) $\mathrm{Re}_{2} \mathrm{O}_{8}$ cube

Metal-oxygen distances

|  | Uncorrected | Corrected |
| :---: | :---: | :---: |
| $\mathrm{Re}(1)-\mathrm{O}(2,1)$ |  |  |
| $\mathrm{O}(2,2)$ | $1 \cdot 915$ (3) | 1.917 |
| $\mathrm{O}(2,5)$ $\mathrm{O}(2,6)$ |  |  |
| O(2,3) | $3 \cdot 266$ (3) | $3 \cdot 267$ |
| $\mathrm{O}(2,4)$ |  |  |
| $\bigcirc(2,7)$ |  |  |
| $\mathrm{O}(2,8)$ |  |  |
| Metal-metal separations |  |  |
| $\operatorname{Re}(1)-\operatorname{Re}(2)$ | $2 \cdot 259$ (1) | $2 \cdot 262$ |
| $\operatorname{Re}(2)^{\prime}$ | $3 \cdot 752$ (1) | $3 \cdot 754$ |
| $\mathrm{La}(5)$ | 3.538 (1) | $3 \cdot 541$ |
| $\mathrm{La}(6)$ |  |  |
| La(7) |  |  |

Bond angles
Distances between the O atoms

| $\mathrm{O}(2,1)-\mathrm{Re}(1)-\mathrm{O}(2,5)$ | $87 \cdot 2$ (1) | 2.642 (4) | $2 \cdot 648$ |
| :---: | :---: | :---: | :---: |
| $\mathrm{O}(2,6)$ |  |  |  |
| $\begin{array}{ll}\mathrm{O}(2,2) & \mathrm{O}(2,5) \\ & \mathrm{O}(2,6)\end{array}$ |  |  |  |
| $\mathrm{O}(2,1) \quad \mathrm{O}(2,2)$ | 154•7 (2) | 3.736 (5) | $3 \cdot 741$ |
| $\begin{array}{lr}\mathrm{O}(2,5) & \mathrm{O}(2,6) \\ \mathrm{Re}(2)-\mathrm{Re}(1)-\mathrm{O}(2,1)\end{array}$ | 154.7 (2) | 3.736 (5) |  |
| $\mathrm{Re}(2)-\mathrm{Re}(1)-\mathrm{O}(2,1)$ $\mathbf{O}(2,2)$ $\mathbf{O}(2,5)$ | $102 \cdot 7$ (1) |  |  |
| $O(2,5)$ $O(2,6)$ | 102.7 (1) |  |  |

Remaining oxygen distances

(b) $\mathrm{LaO}_{8}$ cube

Metal-oxygen distances
Uncorrected Corrected

| $\left.\begin{array}{c}\mathrm{La}(1)-\mathrm{O}(1,2) \\ \mathrm{O}(1,3) \\ \mathrm{O}(2,1) \\ \mathrm{O}(2,3)\end{array}\right\}$ | $2.406(1)$ | 2.406 |
| :---: | :---: | :---: |
| $\left.\begin{array}{c}\mathrm{O}(2,10) \\ \mathrm{O}(2,12) \\ \mathrm{O}(2,13) \\ \mathrm{O}(2,15)\end{array}\right\}$ | $2.537(3)$ | 2.538 |
| $\left.\begin{array}{c}\text { Metal-metal separations } \\ \mathrm{La}(1)-\mathrm{Re}(3) \\ \mathrm{Re}(4) \\ \mathrm{La}(2)\end{array}\right\}$ | $2.571(3)$ | 2.572 |
|  | $2.600(3)$ | 2.600 |
|  | $3.538(1)$ | 3.541 |
|  |  |  |

Table 3 (cont.)

| $\mathrm{O}(2,10)-\mathrm{La}(1)-\mathrm{O}(2,13)$ |  | Bond angles | Distances between the O atoms |  |
| :---: | :---: | :---: | :---: | :---: |
|  |  | $61 \cdot 5$ (1) | $2 \cdot 642$ (4) | 2.648 |
| $\mathrm{O}(2,12)$ | O( 2,15 ) $\}$ | 61.5 (1) |  |  |
| $O(2,13)$ | O $(2,15)$ | $68 \cdot 1$ (1) | $2 \cdot 912$ (7) | $2 \cdot 916$ |
| $O(2,10)$ | $\mathrm{O}(2,12)$ | 69.0 (1) | $2 \cdot 912$ (7) | 2.916 |
| $\mathrm{O}(1,2)$ | $\bigcirc(2,15)$ ) | $71 \cdot 6$ (1) | $2 \cdot 931$ (3) | $2 \cdot 935$ |
| $\mathrm{O}(1,3)$ | $\mathrm{O}(2,13)$ | 71.6 (1) | 2.91 (3) |  |
| $\mathrm{O}(2,3)$ $\mathrm{O}(1,2)$ | $\left.\begin{array}{l} O(2,10) \\ O(2,1) \end{array}\right\}$ | $72 \cdot 7$ (1) | $2 \cdot 931$ (3) | 2.935 |
| $\mathrm{O}(1,3)$ | O $(2,3)$ \} | 72.7 (1) | 2.931 (3) |  |
| $\mathrm{O}(2,1)$ | $\mathrm{O}(2,3)$ | 75.3 (1) | 3.099 (7) | $3 \cdot 103$ |
| $\mathrm{O}(1,2)$ | $\mathrm{O}(1,3)$ | 77.3 (1) | 3.005 (1) | 3.010 |
| $\mathrm{O}(2,10)$ | $\mathrm{O}(2,15)$ | 99.0 (1) | 3.932 (5) |  |
| $\mathrm{O}(2,12)$ $\mathrm{O}(1,2)$ | $O(2,13)$ $O(2,12)$ | (1) | 3.932 (5) |  |
| $\mathrm{O}(1,3)$ | $\mathrm{O}(2,10)$ | $106 \cdot 2$ (1) | 3.982 (3) |  |
| $\mathrm{O}(2,1)$ | $\mathrm{O}(2,15)$ | 107.6 (1) | $4 \cdot 145$ (4) |  |
| $\mathrm{O}(2,3)$ $\mathrm{O}(2,1)$ | $O(2,13)$ $O(2,10)$ | 112.3 (1) | 4.241 (4) |  |
| O( 2,3 ) | $\mathrm{O}(2,12)$ | $112 \cdot 3$ (1) | $4 \cdot 241$ (4) |  |
| $\mathrm{O}(1,2)$ | $\mathrm{O}(2,13)$ | $112 \cdot 5$ (1) | $4 \cdot 164$ (3) |  |
| $\mathrm{O}(1,3)$ | $\bigcirc(2,15)$ | 112.5 (1) | 4164 (3) |  |
| $\mathrm{O}(1,2)$ $\mathrm{O}(1,3)$ | $\left.\begin{array}{l}\mathbf{O}(2,3) \\ \mathbf{O}(2,1)\end{array}\right\}$ | 117.7 (1) | $4 \cdot 231$ (3) |  |
| O(1,2) | $\bigcirc \mathrm{O}(2,10)$ |  |  |  |
| $\mathrm{O}(1,3)$ | $\mathrm{O}(2,12)$ | $170 \cdot 5$ (1) | $4 \cdot 960$ (3) |  |
| $\mathrm{O}(2,1)$ | $\bigcirc(2,13)$ | $170 \cdot 6$ (1) | $5 \cdot 119$ (3) |  |
| O( 2,3 ) | O( 2,15 ) | 1706 (1) | S119 (3) |  |

sentations of the thermal vibrations of the atoms. Corrections for the thermal motions have been applied to the $\mathrm{Re}-\mathrm{O}$ and $\mathrm{La}-\mathrm{O}$ distances, based on the riding-motion model. The metal-metal and oxygen-oxygen distances are corrected under the assumption of independent movements. The most important distances and angles are given in Table 3.

A final three-dimensional difference-Fourier synthesis was computed in sections $0 \cdot 30 \AA$ along the $c$ axis. The residual peaks showed a maximum height of less than $14 \%$ of the smallest oxygen peak in the corresponding ( $F_{o}-F_{\text {metal }}$ ) synthesis.


Fig. 3. $\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$ structure projected on (001). The two sorts of fluorite subcells are outlined with dotted lines.

## Description of the crystal structure

The strusture of $\mathrm{La}_{2} \mathrm{ReO}_{5}$, projected on (001) in Fig. 3, is closely related to the fluorite type. The unit cell contains four Re , eight La , and twenty O atoms. The atoms are numbered according to the system set forth in Table $1(a)$.

All the O atoms, forming a somewhat distorted primitive cubic oxygen arrangement, are in the fluoride positions, while four of the five Ca sites are occupied by La atoms and the fifth accommodates a Re atom pair oriented parallel to the $c$ axis. Because of the Re doublets, its seems advisable to write the formula as $\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$ (cf. Fig. 1).

The rhenium-non-metal coordination (cf. Fig. 2a) consists of two parallel square $\mathrm{ReO}_{4}$ units, which form square pyramids with the Re atoms at the tops. The two $\mathrm{ReO}_{4}$ halves are connected by a $\mathrm{Re}-\mathrm{Re}$ bond in an eclipsed conformation. The Re-O distances are $1.915 \AA$ and the Re atom is $0.42 \AA$ above the plane of the square pyramid. Two $\mathrm{ReO}_{4}$ square pyramids, regarded as a unit $\mathrm{Re}_{2} \mathrm{O}_{8}$, form an elongated cube with two kinds of $\mathrm{O}-\mathrm{O}$ distances, viz. eight distances of 2.642 and four of $3.099 \AA$ (average $2.794 \AA$ ) (see Table $3 a$ ). The Re-Re distance is 2.259 (1) $\AA$, which agrees well with the short distance $(2 \cdot 24 \AA)$ found in the $\mathrm{Re}_{2} \mathrm{Cl}_{8}^{2-}$ ion (Cotton \& Harris, 1965; see also Kuznetzov \& Koz’min, 1963), with the conformation similar to the $\mathrm{Re}_{2} \mathrm{O}_{8}$ unit.

The eight $\mathrm{La}-\mathrm{O}$ distances range from $2 \cdot 406$ to $2 \cdot 600$ $\AA$ with the mean value $2.53 \AA$ (see Table 3), comparable with those found in $\mathrm{La}_{4} \mathrm{Re}_{6} \mathrm{O}_{19}$. The La coordination polyhedron is illustrated in Fig. 2(b).

Each $\mathrm{O}(1)$ atom is coordinated to four La atoms to form a perfect tetrahedron, while the $\mathrm{O}(2)$ atom is bonded to one Re and three La atoms in a rather distorted tetrahedral arrangement. As a result of this difference in function of the two O atoms, a significant shortening of the La-O(1) distance ( $2 \cdot 406 \AA$ ) compared to the $\mathrm{La}-\mathrm{O}(2)$ bond lengths $(2 \cdot 537-2 \cdot 600 \AA)$ is observed, resulting in a deformed cubic environment of the lanthanums.

It is interesting to compare the structure of $\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$ with $\mathrm{Nd}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{11}$ (Wilhelmi et al., 1970) and $\mathrm{La}_{4} \mathrm{Re}_{6} \mathrm{O}_{19}$ (Morrow \& Katz, 1968; Longo \& Sleight, 1968). The structures of $\mathrm{Nd}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{11}$ and $\mathrm{La}_{4} \mathrm{Re}_{6} \mathrm{O}_{19}$ both contain double octahedra $\mathrm{Re}_{2} \mathrm{O}_{10}$ occurring in edge-shared pairs with short $\mathrm{Re}-\mathrm{Re}$ contacts ( $2.42 \AA$ ). In $\mathrm{La}_{4} \mathrm{Re}_{6} \mathrm{O}_{19}$ the dimeric units $\mathrm{Re}_{2} \mathrm{O}_{10}$ are further linked by sharing corners to form a threedimensional network in which there are voids containing $\mathrm{La}_{4} \mathrm{O}$ tetrahedra. The La-O coordination is a trigonal prism which is capped on all three rectangular
faces and one triangular face. In $\mathrm{Nd}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{11}$ the double octahedra are joined together by two crystallographically different neodymiums. One half of the Nd atoms, together with the $\mathbf{O}$ atoms, are in a somewhat deformed fluorite arrangement. The environment around the remaining Nd atoms is a bicapped trigonal prism.
The number of $d$ electrons per Re in $\mathrm{Nd}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{11}$ and $\mathrm{La}_{4} \mathrm{Re}_{6} \mathrm{O}_{19}$ is 2 and $2 \frac{2}{3}$ respectively. Extended Hückel calculations (Sleight, Hare \& Sleight, 1968) showed the presence of a $\mathrm{Re}=\mathrm{Re}$ double bond across the edge-shared octahedra. The significantly shorter Re-Re bond length of $2.26 \AA$, observed in $\mathrm{La}_{4}\left[\mathrm{Re}_{2}\right] \mathrm{O}_{10}$ with three $d$ electrons per Re , indicates that the bonding conditions are more similar to those in the $\mathrm{Re}_{2} \mathrm{Cl}_{8}^{2-}$ cluster (five $d^{4}$ configuration) with an estimated quadruple bond (Cotton, 1965).

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